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A. Lahmar, M. Jouiad, M. El Marssi, H. Mezzourh, S. Ben Moumen, et al.. Effect of growth time on structural and surface properties of TiO₂ nanostructures deposited by single-step hydrothermal method. MATERIALS TODAY-PROCEEDINGS, 2022, 51 (6, SI), pp.2053-2058. 10.1016/j.matpr.2021.08.004 . hal-03634267

HAL Id: hal-03634267

<https://u-picardie.hal.science/hal-03634267>

Submitted on 22 Jul 2024

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Effect of growth time on structural and surface properties of TiO₂ nanostructures deposited by single-step hydrothermal method

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Abstract

In this study, highly pure rutile TiO₂ nanorods (NRs) structures were grown on fluorine-doped tin oxide (FTO) coated glass substrates by using a one-step hydrothermal method. The effect of the growth time on the structure and surface properties of the obtained products has been studied. The X-ray diffraction analysis and the Raman spectroscopy indicated that the films were single-crystalline rutile phase. At lower hydrothermal time, TiO₂ NRs singly distributed with small diameter are found showing a high contact angle with ethylene glycol. When the hydrothermal reaction increases, nanoflowers (NFs) with a large diameter are formed over the TiO₂ NRs leading to superhydrophilic surfaces. These nanostructured thin films can be tailored to the requirements of specific applications involving water and ethylene glycol.

Keywords: TiO₂ nanorods/nanoflowers; hydrothermal; FTO/TiO₂; contact angle; superhydrophilicity; wettability.

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1. Introduction

Wettability and superhydrophilicity are some of the surface properties, which play an important role in a broad range of practical applications, including self-cleaning and anti-fogging, wear and erosion-resistant defensive, transportation of fluids in micro-fluidic devices, and development of bio-medical materials [1]. In general, surface wettability can be strongly controlled by surface chemistry, free energy, morphology, and liquid nature[2]. A facile way to express superwetting characteristics of a surface is by the contact angle (CA) of liquid droplets resting on the surface. Wettability studies require the measurement of contact angles as the primary data, which indicates the degree of wetting at the time of solid-liquid interaction[3].

One-dimensional (1D) nanostructures (e.g., nanotubes, nanobelts, nanowires, nanorods, and nanoflowers) have been considered to be very attractive candidates for superhydrophilic (superwetting) surfaces fabrication due to their surface roughening[4]. Particularly, single-crystalline rutile TiO₂ nanostructures are considered to be favorable surface wettability materials because of their excellent chemical stability, structure controllability, specific interface effect, and low cost[5]. The superhydrophilic properties of TiO₂ nanostructures can be brought by many technologies, including plasma treatment, ultraviolet irradiation, laser treatment, ultrasonic spray pyrolysis, sol-gel process, anodization, electrospinning, laser deposition, microwave irradiation, and hydrothermal treatment [1].

In this paper, we report a simple and fast growth of aligned TiO₂ nanostructures on FTO substrates by a one-step hydrothermal method. The influence of growth time on structure and surface properties was discussed.

2. Experimental procedure

2.1. Chemical and materials

All chemicals were used without any further treatment. Titanium (IV) isopropoxide (TTIP) with 98% purity was purchased from Sigma-Aldrich and used as a precursor. Hydrochloric acid (HCl) with 37% concentration was supplied by Labbox and used as a catalyst. Deionized (DI) water was obtained from local sources and FTO glass substrates from MSE Supplies.

2.2. Synthesis procedure

2.2.1. Substrate preparation

The substrates were ultrasonically cleaned in acetone, propan-2-ol, and absolute ethanol for 30 min followed by deionized water. Then the substrates were dried under N₂ gas.

2.2.2. Preparation of aligned TiO₂ nanorods (NRs)

The synthesis of the TiO₂ nanorods (NRs) was performed by a single-step hydrothermal method under relatively acidic conditions on a transparent FTO-coated glass substrate. In a typical preparation, 10 mL of deionized water was mixed with 10 mL of concentrated hydrochloric acid solution to reach a total volume of 20 mL, and then the mixture was stirred for 10 min. A certain amount of titanium (IV) isopropoxide was added to the previous mixture solution to offer a sufficient titanium source and then stirred for 30 min. The final solution was transferred into a Teflon-lined stainless steel autoclave with a volume ratio of 2/3. Meanwhile, a clean piece of the FTO glass substrate was placed at the bottom of the autoclave. Then, the hydrothermal reaction was conducted at 180 °C for different reaction times at 1, 1.5, 2, and 3 h. The corresponding samples are designated TO-x, where x is the growth time.

After the reaction time is over, the autoclave was cooled naturally to room temperature. The obtained samples were repeatedly rinsed with DI water and ethanol to remove impurities and finally dried with N₂ gas. Finally, the dried samples were annealed at 350 °C for 1h.

2.3. Technical characterization

Structural characterization of the prepared TiO₂ nanostructures was determined by X-ray diffraction (XRD, Panalytical X-Pert Pro) analysis using the Cu-K α radiation with $\lambda \sim 1.540598$ Å. The measurements were performed at a range angle from 20° to 80° with a step size of 0.02°. Morphological and compositional characterizations were examined by a Scanning Electron Microscope (SEM, VEGA 3-Tscan) in association with energy dispersive X-ray (EDX) analysis. Contact angles were measured by defining a circle around the liquid drops (deionized water and ethylene glycol (EG) and recording the tangent angle formed at the substrate surface. The droplet volume used was 5 μ L.

3. Results and discussion

3.1. Structural analysis

Figure 1 shows XRD patterns for the FTO/TiO₂ nanostructures prepared at different reaction time intervals (1, 1.5, 2, and 3 hours) and the FTO substrate. All the diffraction peaks can be indexed as the tetragonal rutile structure of TiO₂ (JCPDS card number 89-4920). The diffraction peaks at 2θ of 26.88, 34.09, 38.11, 51.83, 61.89, and 65.79 are assigned to tin oxide (SnO₂) arising from FTO substrate (JCPDS card no. 41-1445). No impurity peaks have been detected from the XRD spectrums. By increasing the growth time, the peak intensities for the TiO₂ rutile phase increase at the expense of FTO peak intensities which could be explained by the increase in the thickness of the TiO₂ film. The diffraction pattern of TO-1h shows only one peak of rutile (211) at 2θ of 54.49°, while the film synthesized with a growth time of 1.5h reveals four peaks of rutile (101), (211), (002), and (301). The TO-2h and TO-3h show clearly peaks of the rutile phase, which is due to the fact that almost the entire surface of FTO has been completely covered with TiO₂ rutile-phased nanostructures. It should be noted that by increasing the growth time, the (002) peak gets more intense, suggesting that the degree of crystallinity and the crystallographic orientation growth along the [001] direction become more pronounced, resulting in the anisotropic growth of rutile TiO₂ nanorods along the [001] direction. The average crystallite size was estimated from XRD patterns using the Scherrer equation, and the found values are gathered in **Table 1**. It is observed that the average crystallite size of TiO₂ nanostructures increases by increasing the growth time, indicating better crystallinity for a longer hydrothermal reaction time.

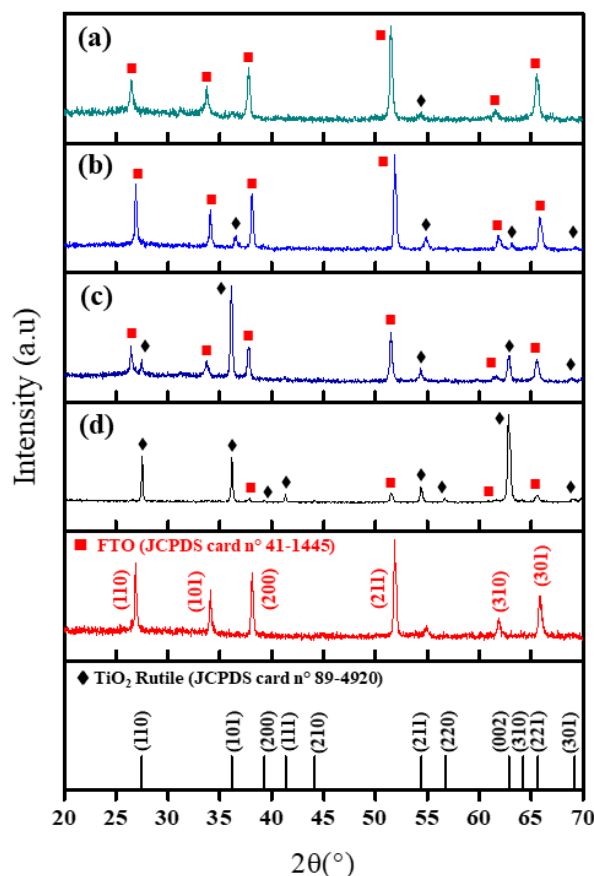


Fig.1. XRD patterns for TiO₂ nanostructures prepared via hydrothermal technique with different reaction times: (a) 1 h, (b) 1.5 h, (c) 2 h, and (d) 3 h.

Table.1. Average diameter size, FWHM, and average crystalline size of TiO₂ nanostructures prepared at different reaction times.

Reaction time (h)	Average diameter (nm) by SEM		FWHM	Average crystalline size (nm) by XRD
	NRs	NFs		
1	49.80	---	0.5510	16.22
1.5	114.06	111.86	0.3149	28.43
2	125.85	173.83	0.1574	56.76
3	250.35	582.08	0.0787	113.53

3.2. Raman spectroscopy

To evaluate the structural changes of the synthesized TiO₂ nanostructures, Raman spectroscopy was also performed. **Figure 2** shows the Raman spectra of TiO₂ nanostructures at different reaction times (1, 1.5, 2, and 3h). The spectra reveal that no peaks other than those pertaining to rutile TiO₂ appear. It can be seen from the four spectra that the peak intensities are significantly increased with increasing the reaction time, which is in good accordance with the XRD results. After 3h of reaction, it is clearly seen that the first-order Raman scattering of rutile TiO₂ showed active fundamental modes at 149.0 cm⁻¹ (B_{1g}), 234.9 cm⁻¹, 449.5 cm⁻¹ (E_g), 613.3cm⁻¹ (A_{1g}), and 826.5 cm⁻¹ (B_{2g}). The two predominant maxima peaks located at 449.5 cm⁻¹ and 613.3cm⁻¹ correspond to the O–Ti–O bending vibrations and

Ti–O stretching vibrations of crystalline rutile TiO₂ phase, respectively [6]. Besides, there is a second-order scattering featured at about 234.9 cm⁻¹ peak due to the multiple-phonon scattering process, which is also considered as a characteristic Raman peak of rutile TiO₂.

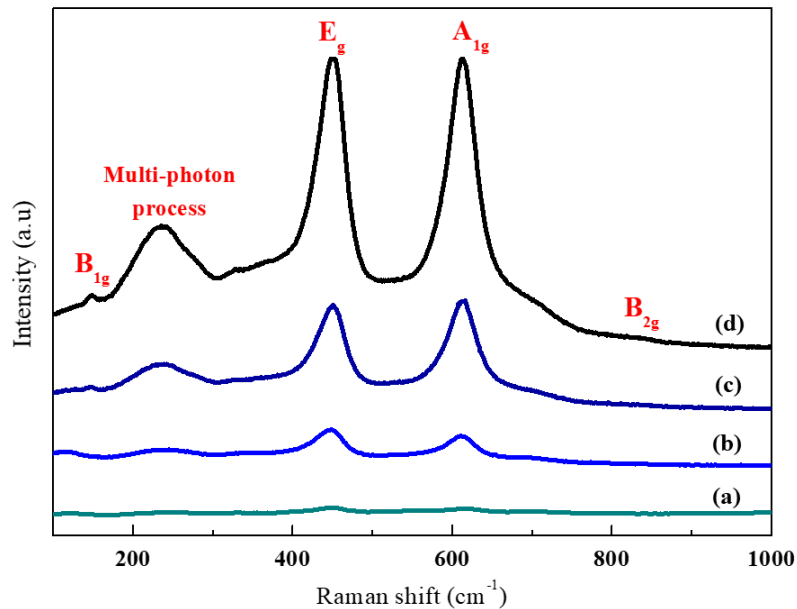


Fig.2. Raman spectra of samples (a) TO-1h, (b) TO-1.5h, (c) TO-2h, and (d) TO-3h.

3.3. Microstructure and compositional analyses

The surfaces' morphologies of the bare FTO substrate and FTO/TiO₂ nanostructures are observed and the results are shown in **Figure 3(a–f)**. The SEM images revealed that the TiO₂ nanostructures with different densities, sizes, and shapes can be achieved by increasing the hydrothermal reaction time. At 1h of reaction time, the shapes of the structures were found to be nanorods singly distributed on FTO substrate with of diameter of about 50 nm (**Figure3b**). As the reaction time becomes longer, a significant increase in dimension of the resultant nanostructures is observed. The nanorods become larger and the gap between them decreases. Likewise, 3D nanoflowers (NFs) like arrangements of NRs are formed over the 1D nanorods as shown in **Figure3(b–e)**. As we can see in the zoom of the SEM micrograph in **Figure 3f**, the NRs developed on the TO-3h NRs surface reveal a regular tetragonal shape with smooth walls and square end, which is a signature of rutile crystal structure in accordance with XRD measurements [7]. Furthermore, the average size of NRs and NFs increases with increasing hydrothermal reaction time (**Table 1**). It is apparent from this study that the hydrothermal reaction time has a significant effect on the TiO₂ NRs microstructure. Indeed, aligned TiO₂ NRs are found when the reaction time is less than 1.5h, but beyond this growth time, NFs become to grow over the NRs coating. According to the Wenzel theory[8], these interesting roughness hierarchical TiO₂ NRs structure would be a benefit to enhance the wettability of surfaces. **Figure 4** shows the schematic illustration of the growth mechanism of TiO₂ nanorods on FTO substrate.

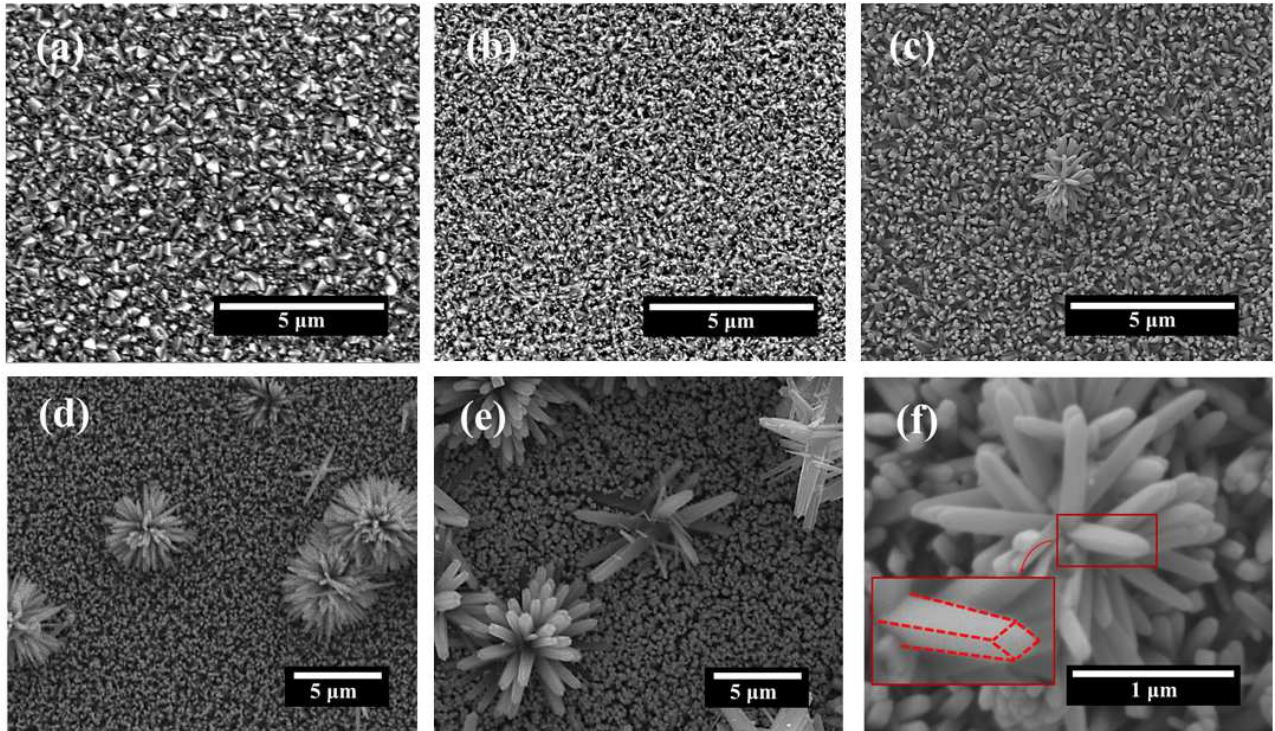


Fig.3. SEM images of a bare FTO substrate (a) and TiO₂ nanostructures prepared via hydrothermal technique with different reaction times: (b) 1 h, (c) 1.5 h, (d) 2 h, and (e) 3 h. (f) zoom of the NF growth on TO-1.5h NRs.

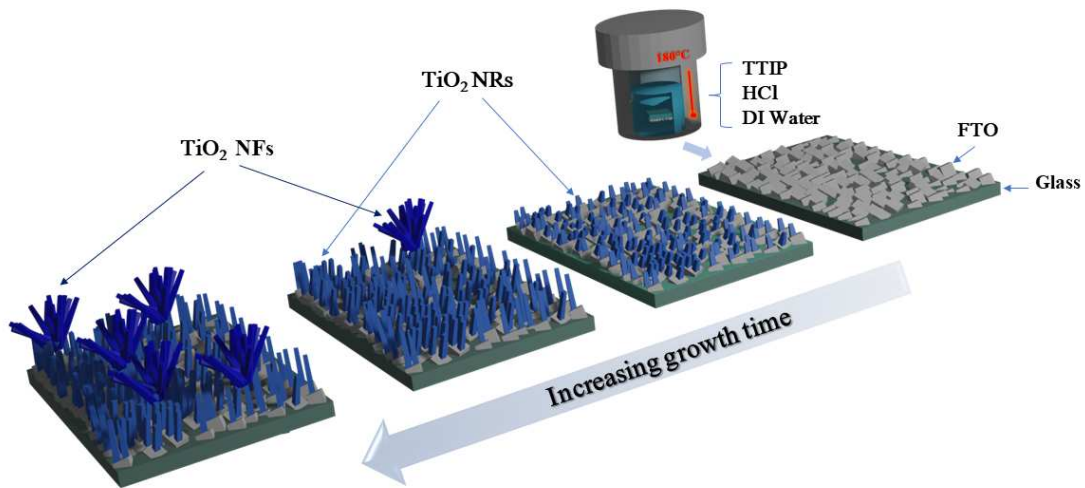


Fig.4. Scheme for growth mechanism of TiO₂ nanostructures on FTO substrate.

The elemental composition of the TO-1.5h sample was analyzed by EDX as shown in **Figure 5**, there is no trace of any other impurities that could be seen within the detection limit of the EDX. The corresponding EDX result indicates that the TO-1.5h film consists mainly of Ti and O with an approximate atomic ratio of 1:2 which is in consistent with the stoichiometry of TiO₂. Hence, EDX analysis confirms the formation of TiO₂.

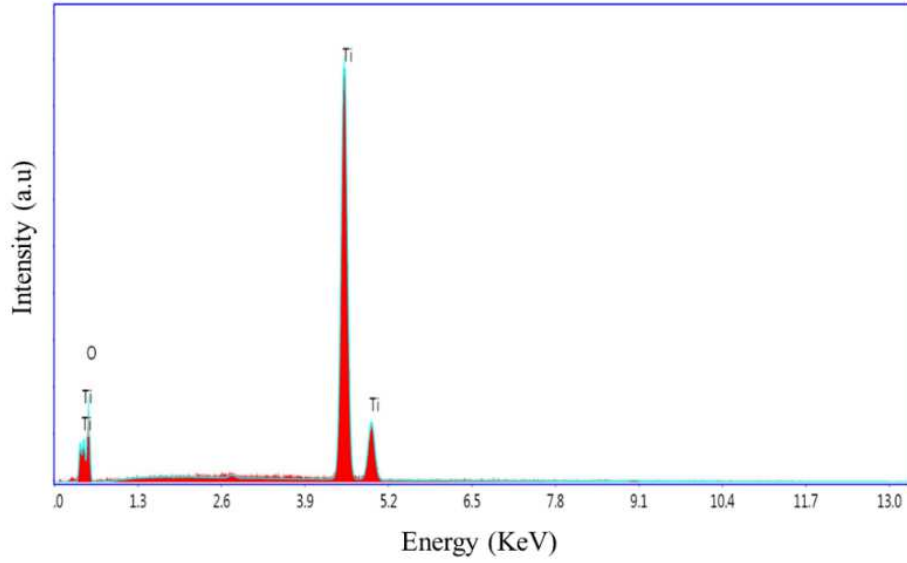


Fig.5. EDX spectrum of TiO₂ nanostructures prepared at 1.5 h reaction time.

3.4. Wetting properties

The interfacial reaction of TiO₂ nanostructures could be followed by measuring the contact angle. According to the Young's equation [9], the contact angle θ of a droplet on the surface can be described as:

$$\cos \theta_e = \frac{\gamma_{SV} - \gamma_{SL}}{\gamma_{LV}} \quad (1)$$

Where θ_e is the equilibrium contact angle. γ_{SV} , γ_{SL} , and γ_{LV} , are the interface energies at the solid-vapor, solid-liquid, and liquid-vapor interfaces, respectively.

Depending on the contact angle obtained, we can define the characteristics of the observed surfaces. If the resulting contact angle is 0°, the surface is mentioned as superhydrophilic and if it is less than 90°, the surface is hydrophilic. If the CA is between 90 and 180°, the surface is hydrophobic and if the angle is greater than 180°, the surface is superhydrophobic.

It should be noted that the wetting behavior of a solid surface depends not only on the properties of the liquid and the chemical properties of the solid surface but also on its roughness. Thus, to create superhydrophilicity, the surface roughness must be controlled to maximize the contact area and interaction between liquid and the solid surface. Wenzel and Cassie-Baxter suggested two models to describe the wetting phenomena on rough surfaces. According to Wenzel's model, the water droplet seeps in between the irregularities of a rough surface increasing the contact area between the solid surface and the liquid. On the other hand, for the Cassie-Baxter model, the water droplet sits on top of the surface structure without much penetration [10], [11]. The rougher the surface, the higher will be the contact angle.

To investigate the wettability property of the TiO₂ nanostructures, the water and ethylene glycol contact angle measurements were carried out. **Figure 6** shows the change in the contact angle of TiO₂ nanostructures at various growth times. The TiO₂ surfaces were completely wetted by water and exhibited almost zero water contact angles (WCA). This superhydrophilicity could be attributed to the rough TiO₂ NRs surface and to the presence of hydroxyl groups on its surfaces [12]. Because of the capillary effects of the rough surface, the water droplet can be induced to

enter into the NRs and fill the grooves of the coating according to the model proposed by Wenzel [8]. The TiO₂ nanostructures show a good wetting surfaces compared with other TiO₂ films reported in the literature (see **Table 2**). Furthermore, these nanostructures present a perfect wetting similar to that reported by W. Yang et al. and R. Dong.

In the case of ethylene glycol (EG), widely used as coolant and antifreeze agent in many commercial and industrial applications [13] the as-prepared TiO₂ nanostructures at 1h have produced a large contact angle of 83.3°. It is important to highlight that higher CA of ethylene glycol is very useful for its application as a corrosion inhibitor in cleaning processes [13]. When the growth time increases, the EG contact angle decreases abruptly to a very small value of CA about 5° for TO-3h, showing a good surface wettability. This trend can be explained by the formation of a highly porous TiO₂ structure to increase the surface roughness to achieve superhydrophilic surfaces as it has been noted by several authors [14], [15].

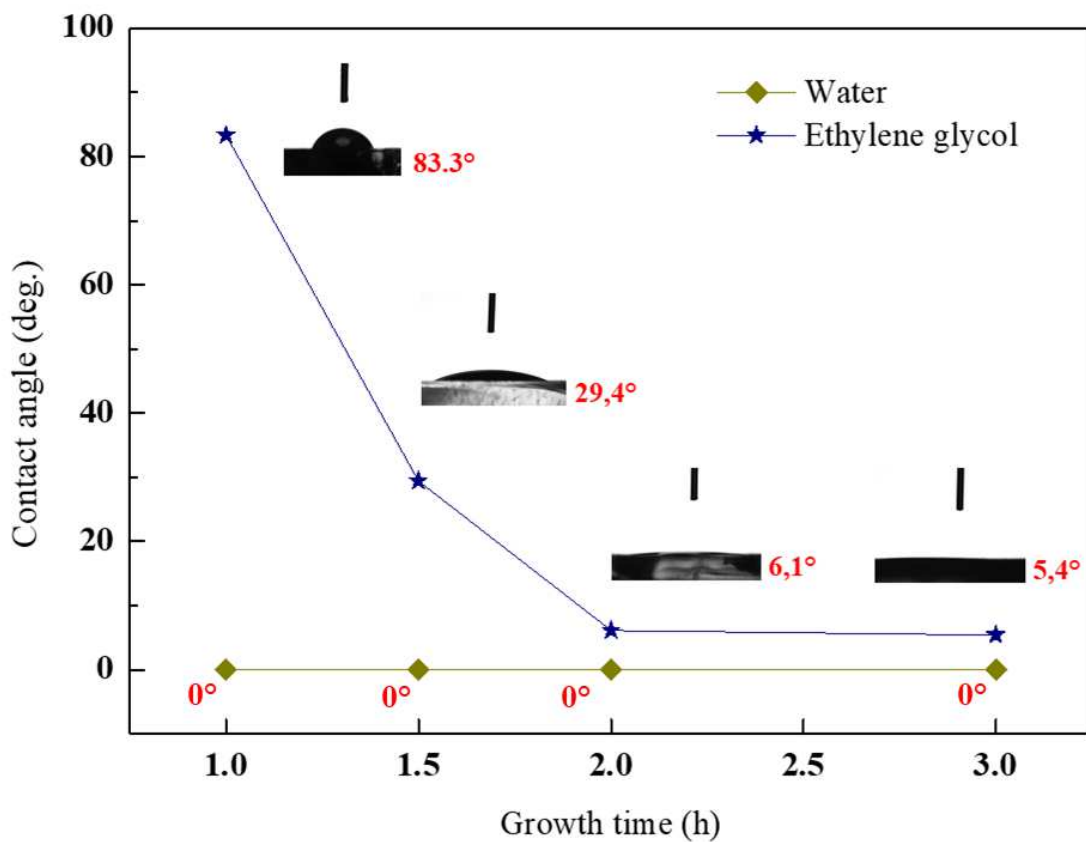


Fig.6. Contact angle properties of TiO₂ nanostructures at different growth times.

Table.2. Comparison of the contact angle properties of TiO₂ nanostructures prepared at different reaction times with other TiO₂ films reported in the literature.

Sample	Morphology	Diameter (nm)	Solvent	Contact angle (°)	Illuminating source	Surface wettability	Refs
TO-1h nanostructured film	Nanorods	49.80	Water	0	Visible light	Perfect wetting	Our work
	Nanoflowres		Ethylene glycol	83.3		High wetting	
TO-1.5h nanostructured film	Nanorods	114.06	Water	0	Visible light	Perfect wetting	Our work
	Nanoflowres		Ethylene glycol	29.4		High wetting	
TO-2h nanostructured film	Nanorods	125.85	Water	0	Visible light	Perfect wetting	Our work
	Nanoflowres		Ethylene glycol	6.1		High wetting	
TO-3h nanostructured film	Nanorods	250.35	Water	0	Visible light	Perfect wetting	Our work
	Nanoflowres		Ethylene glycol	5.4		High wetting	
TiO ₂ nanotube arrays	Nanotubes	Not specified	Water	0	UV illumination	Perfect wetting	[16]
TiO ₂ nanorod film	Nanorods	250	Water	0	Visible light	Perfect wetting	[17]
TiO ₂ nanosheet array films	Nanosheets	30-70	Water	3	UV illumination	High wetting	[18]
TiO ₂ film	Nanoparticles	100–200	Water	< 10	Visible light	High wetting	[19]
TiO ₂ film	Nanoparticles	50	Water	≈ 30	UV illumination	High wetting	[20]
TiO ₂ nanorods	Nanorods	97 - 120	Water	31 - 37	Not specified	High wetting	[21]

4. Conclusion

Superhydrophilic rutile TiO₂ Nanorod Arrays surfaces covered with NFs-like structure have been successfully fabricated on the FTO coated glass deposited by a single-step hydrothermal method. The CA on the as-deposited surfaces can reach 0° for water and 5° for EG droplets, showing outstanding superhydrophilicity. The formation of more nanoflowers with a large diameter on the top of the nanorods layer improves the surface wettability; whereas nanostructured film with smaller nanorods formed at a low hydrothermal time produced a higher contact angle of 83° with EG. These findings are of crucial interest for the design of self-cleaning surfaces and erosion-resistant defensive.

ACKNOWLEDGMENTS:

This research was supported by CNRST Priority Program PPR 15/2015 and the European H2020-MSCA-RISE-2017-ENGIMA action.

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